

# Translocation of structured polynucleotides through nanopores

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We investigate theoretically the translocation of structured RNA/DNA molecules through narrow pores which allow single but not double strands to pass. The unzipping of basepaired regions within the molecules presents significant kinetic barriers for the translocation process. We show that this circumstance may be exploited to determine the full basepairing pattern of polynucleotides, including RNA pseudoknots. The crucial requirement is that the translocation dynamics (i.e., the length of the translocated molecular segment) needs to be recorded as a function of time with a spatial resolution of a few nucleotides. This could be achieved, for instance, by applying a mechanical driving force for translocation and recording force-extension curves (FEC's) with a device such as an atomic force microscope or optical tweezers. Our analysis suggests that with this added spatial resolution, nanopores could be transformed into a powerful experimental tool to study the folding of nucleic acids.

A series of recent experiments studied the translocation of DNA and RNA molecules through narrow pores, which allow single but not double strands to pass [1–8]. These investigations pursued two main goals: (i) to probe in a well-defined model system the physics of biopolymer translocation across membranes, a process which is ubiquitous in cell biology, and (ii) to explore the potential of nanopores as a single-molecule tool. In the experiments so far, a membrane protein,  $\alpha$ -hemolysin, was used as the pore. An electric field acting on the negatively charged DNA/RNA backbone drives the molecules through the pore, and translocation is monitored by measuring the induced ionic current, which is strongly reduced while a DNA/RNA chain blocks the pore. Until very recently [5, 8], the experiments have focused on the translocation of unstructured, mostly homopolymeric molecules, a problem which has also received considerable theoretical interest [9–15]. For such unstructured molecules, the main results regarding the above goals were that (i) the basic physics of translocation is adequately described by a drift-diffusion process, in which monomers hop randomly in and out of the pore with a directional bias due to the applied voltage [12], and (ii) nanopores could possibly be developed into rapid sequencing devices, since the ionic current during blockage displays a weak sequence-dependence [2, 3].

In contrast, for structured polynucleotides, both the basic physics and the potential applications of translocation still remain largely unexplored. Experimentally, important first steps have been taken by studying the translocation of simple hairpin (i.e., stem-loop) structures [5] and the unzipping of double-stranded DNA through a nanopore [8]. However, a general theoretical

framework to describe translocation of these as well as more complex RNA/DNA structures is currently lacking. Here, we first construct such a framework and then use it to investigate the potential of nanopores as a single-molecule tool for the study of biopolymer folding.

In this article, we are interested in the generic physical aspects of the translocation process that neither depend on the specific properties of a particular protein pore, nor on the detailed way in which the driving force for translocation is applied. As in previous theoretical studies [9–15], we use a coarse-grained model which treats the pore basically as a separator between a *cis* and a *trans* part of the molecule with a characteristic friction coefficient, see the sketch in Fig. 1. Presumably this description will apply directly to solid-state nanopores [16, 17], which can now be fabricated with a high level of control for the pore geometry, permitting in principle the optimization for different applications. Also, we do not consider the full three-dimensional (tertiary) structure of the molecules, but focus on the basepairing pattern, i.e.

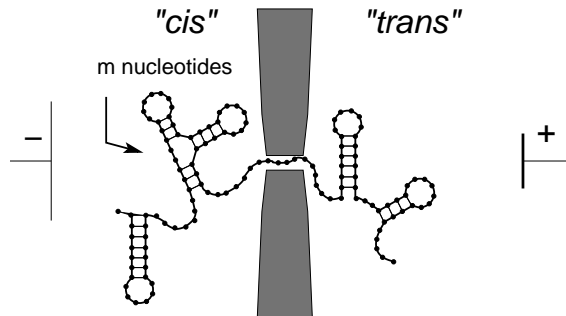


FIG. 1: Sketch of a structured polynucleotide that is driven across a nanopore which allows single but not double strands to pass. Here, the driving force causing translocation from the *cis* to the *trans* side is exerted by an electric field that acts on the negatively charged backbone of the molecule.

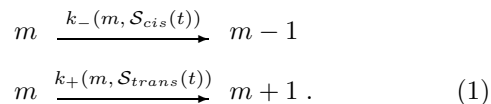
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the secondary structure including possible pseudoknots, which are the only structural features present when there are no divalent metal ions in the solution. Unless stated otherwise, the term ‘structure’ refers here to this base-pairing pattern. While both our theoretical framework and our conclusions apply equally to RNA and single-stranded DNA, the RNA case is particularly interesting, since structured RNA’s have a multitude of functions in molecular biology and RNA folding is an active field of research [18–21].

### General theoretical framework

Fig. 1 depicts schematically the driven translocation of a structured polynucleotide from the *cis* to the *trans* side of the pore. We seek here a convenient reduced description of this translocation process, rather than modeling the full three-dimensional polymer dynamics explicitly. Our approach is similar in spirit to the existing models for the case of unstructured polymers [9, 11, 12], where the translocation dynamics is formulated in terms of a single variable, e.g. the number of nucleotides,  $m$ , on the *cis* side, see Fig. 1. The dynamics,  $m(t)$ , is stochastic and can be described by ‘hopping rates’,  $k_-(m)$  and  $k_+(m)$ , for forward and backward motion of the nucleotide chain through the pore with a stepsize of one monomer. The external force on the molecule leads to an imbalance in the hopping rates,  $k_-(m) > k_+(m)$ , and hence a mean drift towards the *trans* side. For unstructured molecules the one-dimensional description is permissible, if the relaxation of the polymer degrees of freedom on both sides of the pore is faster than the hopping process. This assumption does not hold for arbitrarily long polymers, since the relaxation time increases with the polymer length [12, 14], however for lengths on the order of a thousand bases, the one-dimensional description is adequate under typical experimental conditions [12]. The residual effect of the polymer ends is then only to introduce an entropic barrier for translocation, which leads to a weak  $m$ -dependence of the hopping rates.

For structured molecules, the translocation dynamics is considerably more complicated, since the dynamics of the ‘reaction coordinate’,  $m(t)$ , is then coupled to the dynamics of the basepairing patterns on both sides: the structure on the *cis* side,  $\mathcal{S}_{cis}(t)$ , affects the forward rate, while the structure on the *trans* side,  $\mathcal{S}_{trans}(t)$ , affects the backward rate,



In two limiting cases however, the process can be modeled by a one-dimensional Brownian walk as for unstructured molecules, but with a complex sequence/structure-

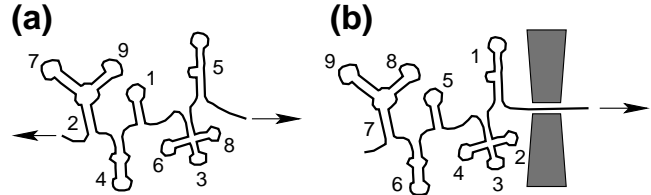


FIG. 2: Unzipping a structured molecule by pulling on its ends is fundamentally different from unzipping by driven translocation through a narrow pore. (a) For pulling on the ends, the stems (i.e., contiguously basepaired segments) in the molecule unfold in an order determined by their relative stability and the topology of the structure (a possible order 1–9 is indicated). (b) In contrast, the pore forces the stems to unfold in a linear order along the sequence, as again indicated by the numbering 1–9.

dependent free energy landscape  $\mathcal{F}(m)$  along the coordinate  $m$ : (A) If the dynamics of the basepairing patterns  $\mathcal{S}_{cis}(t)$  and  $\mathcal{S}_{trans}(t)$  is much faster than the hopping process, the landscape is determined by the ensemble free energy of all basepairing patterns on the *cis* and *trans* side. (B) In the opposite limit, the basepairing pattern on the *cis* side is essentially frozen and is unzipped basepair by basepair as it is driven through the pore. The landscape is then determined by the basepairing energetics of the particular molecular structure prior to translocation, see below. In both cases, the free energy naturally decomposes into three parts,

$$\mathcal{F}(m) = \mathcal{F}_{cis}(m) + \mathcal{F}_{trans}(m) + \mathcal{F}_{ext}(m), \quad (2)$$

where  $\mathcal{F}_{cis}(m)$  and  $\mathcal{F}_{trans}(m)$  denote the intrinsic binding free energies of the *cis* and *trans* parts of the molecule, while  $\mathcal{F}_{ext}(m)$  describes the effect of the external force. Given  $\mathcal{F}(m)$ , the simplest form for the hopping rates  $k_{\pm}(m)$  which satisfies the detailed balance condition  $k_+(m)/k_-(m+1) = e^{-\beta[\mathcal{F}(m+1) - \mathcal{F}(m)]}$  (with  $\beta = 1/k_B T$ ) is

$$k_{\pm}(m) = k_0 e^{-\beta \cdot \max\{\mathcal{F}(m \pm 1) - \mathcal{F}(m), 0\}}. \quad (3)$$

Here,  $k_0$  denotes a microscopic rate constant, which can in principle be tuned by adjusting the properties of the pore. It can be interpreted as a friction coefficient and corresponds approximately to the bare hopping rate for unstructured molecules at zero external force (typical experimental estimates for  $k_0$  are on the order of  $10^5 \text{ s}^{-1}$  [6]). The dynamics of the translocation process, as described by Eqs. (2) and (3) is dominated by energetic barriers due to basepairing, whereas the above-mentioned entropic barrier is completely negligible for structured molecules. These energetic barriers lead to arrests during translocation, as clearly observed already in the experiments with simple hairpins [5] and double-stranded DNA [8].

## Pulling through a pore

**Qualitative aspects.** We now make use of the theoretical framework constructed above to investigate which information on structured molecules could be derived from pore translocation experiments. To this end, it is useful to compare unzipping by driven translocation through a nanopore with the more conventional way of unzipping by applying a force on the ends of a biopolymer, see e.g. [22–24]. As illustrated in Fig. 2, the two approaches differ fundamentally: Pulling on the ends induces a spontaneous unfolding order for the individual structural elements, which is a function of their relative stabilities and the topology of the structure. In contrast, the nanopore prescribes a linear order along the sequence, and unfolds an RNA molecule much like enzymes such as the ribosome do in cells. This difference suggests that the two approaches can also yield different types of information about the molecule under study. As demonstrated by Onoa *et al.* [24], clever use of the pulling on the ends approach can reveal detailed information on the (un)folding pathway of an RNA molecule with known structure. However, when the structure of an RNA molecule is unknown, pulling on the ends can provide, by itself, little information beyond a count of the number of structural elements that unfold separately [24, 25]. In the following we therefore focus on the question of how much structural information may in principle be obtained with the nanopore approach.

Let us suppose that we were able to observe the trajectories  $m(t)$  of the molecules during the translocation process. We could then assign a position within the sequence to each arrest during translocation. Since an arrest is caused by a kinetic barrier, i.e. a stem trapped at the entrance to the pore, we could thereby identify the positions of the stems in the structure. Such information can indeed be sufficient to reconstruct almost the entire basepairing pattern of a molecule, as we demonstrate explicitly using an example below. If the translocation dynamics is in the strongly driven limit (B) where the structure on the *cis* side is essentially frozen, then the reconstructed structure would correspond to the initial structure of the molecule before translocation. We concentrate on this limit in the following, including a discussion of its attainability. However, it may be noteworthy that in the slow translocation limit (A) one would also obtain useful structural information, namely on the average structure of the molecule (with respect to the thermodynamic ensemble of all structures [26]). As long as the molecule is ‘well-designed’ this average structure will be dominated by the ground-state, i.e. the minimum binding free energy structure<sup>1</sup>.

How could one possibly observe the trajectories  $m(t)$  during translocation? For the purpose of structure determination, we will need  $m(t)$  with a spatial resolution below the typical length of a stem in an RNA structure (5–10 basepairs). This may be achievable through a refinement of the current nanopore technology, such that careful analysis of the ionic current allows a count (or even sequencing) of the bases that have passed the pore [2, 3]. With artificial solid-state pores [16, 17] it is also conceivable to use a tunneling current through leads within the membrane as a probe to count (or sequence) the bases as they pass through the pore. Here, we explore yet another option, namely pulling the molecule mechanically through the pore, with a device that can record force-extension curves, e.g. an atomic force microscope or optical tweezers. The explicit discussion of this case with an exemplary RNA sequence serves us to gauge the more general capability of nanopores as single-molecule tools for the study of biopolymer folding.

**Quantitative aspects.** Mechanical unfolding of a biopolymer yields characteristic sawtooth-shaped signatures in the force-extension curve (FEC) indicating the opening of structural elements within the molecule, see e.g. [22, 24]. From the relative positions of these sawteeth one can determine length changes within the molecule with an extremely high resolution of about 1 nm. In the usual setup where the molecule is unfolded by pulling on its ends, such length changes can only be used to infer the ‘stored length’ of a structural element, but not its precise position along the backbone of the molecule, cf. Fig. 2. In contrast, for mechanical pulling through a pore, the relative positions of the resulting sawteeth will correspond directly to the relative positions of the structural elements in the sequence<sup>2</sup>. One conceivable way to prepare the initial condition where an RNA molecule is almost entirely on the *cis* side, with one end threaded through the pore and attached to a pulling device on the *trans* side, is to start with an attached molecule on the *trans* side and to apply a voltage pulse across the pore that suffices to drive the molecule as far as possible to the *cis* side.

To apply our general model to the particular case of

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sponds to the regime where the typical timescale for the translocation of say a single hairpin is comparable to the timescale for structural rearrangements involving the formation of new stems: in this case, the structure on the *cis* side may relax after a stem is unzipped, so that one would observe only the signatures of the relaxed structure rather than the original structure. This regime should be avoided by a proper choice of the driving force and the friction coefficient of the pore (R. Bundschuh and U. Gerland, to be published).

<sup>2</sup> The absolute position can be inferred by adding a known structural element, e.g. a strong C-G hairpin, to one end of the RNA, which can then function as a reference point.

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<sup>1</sup> The worst case for the purpose of structure determination corre-

mechanical pulling in the strongly driven (fast pulling) limit, we need to specify the form of the three terms in the free energy landscape (2). The second term, i.e. the binding free energy on the *trans* side, may be set to zero,

$$\mathcal{F}_{trans}(m) = 0, \quad (4)$$

since the reformation of structure after translocation is suppressed at high tensions in the RNA single strand<sup>3</sup>. The third term,  $\mathcal{F}_{ext}(m)$ , describes the effect of the mechanical stress on the RNA, which stretches the single-stranded *trans* part of the molecule. The elastic response of this single-strand may be modeled by a freely jointed chain (FJC) polymer model. Assuming for simplicity a constant pulling speed  $v$ , the third term then takes the form

$$\mathcal{F}_{ext}(m) = \mathcal{F}_{FJC+spring}(v \cdot t; N - m). \quad (5)$$

Here, the function  $\mathcal{F}_{FJC+spring}(R_t; n)$  denotes the combined free energy of a single-stranded RNA of  $n$  bases in series with a linear spring, stretched to a total extension  $R_t = v \cdot t$  [25]. (The linear spring takes into account the stiffness of the force-measuring device, see the Appendix for details.) By assumption, the first term,  $\mathcal{F}_{cis}(m)$ , represents the binding free energy of the remaining part of the initial structure on the *cis* side.  $\mathcal{F}_{cis}(m)$  can be calculated for any initial structure, based on the free energy rules for RNA secondary structure [27] with a natural extension for pseudoknotted structures, see the Appendix. Our assumption of a frozen structure on the *cis* side is most likely an oversimplification for realistic pulling speeds, since small fluctuations in the secondary structure are known to occur already on timescales on the order of tens of microseconds [28]. However, since the pore pulling approach is sensitive only to stem positions, we expect that it is unaffected by small fluctuations and sensitive only to major rearrangements which significantly change the secondary structure. Such rearrangements are typically slow, sometimes even on the timescale of hours [29, 30].

**Reconstruction of secondary structures.** To illustrate the problem and the method, we use an exemplary RNA, the well-studied self-splicing intron of *Tetrahymena thermophila* [18] with a sequence of 419 bases (Genbank # V01416). In its correctly folded active state, the basepairing pattern of this ribozyme contains a pseudoknot (see Fig. 3a), while its best characterized long-lived folding intermediate [29, 31] has a known alternative structure without pseudoknot (see Fig. 3b) [29].

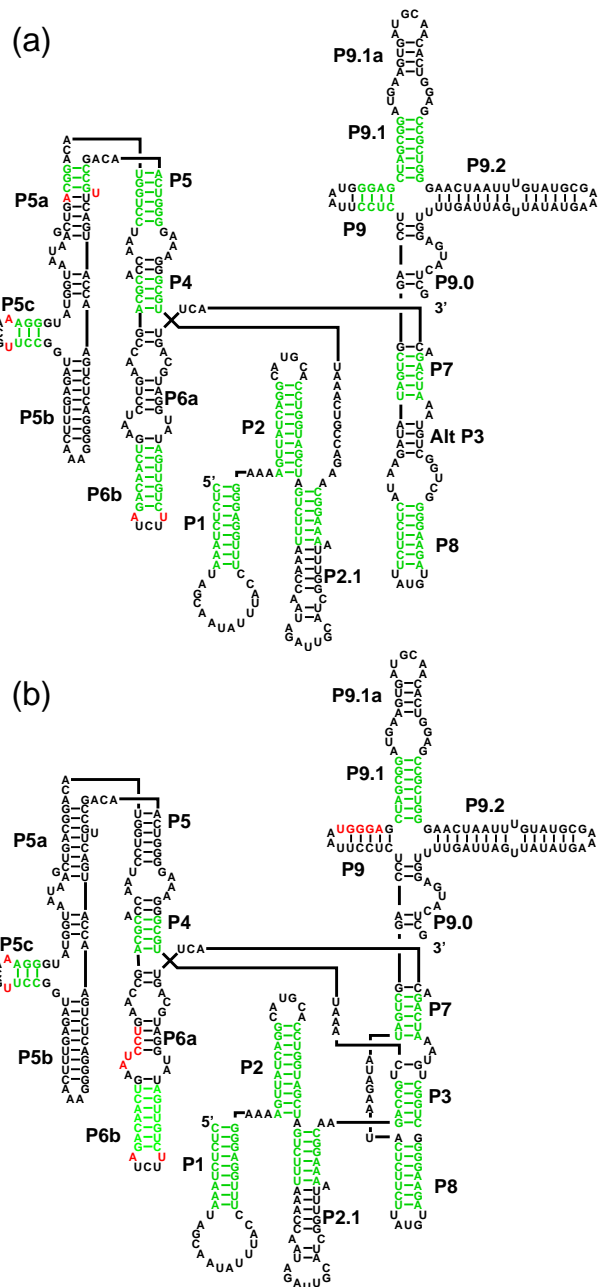


FIG. 3: Secondary structure of the *Tetrahymena thermophila* Group I intron: (a) Long-lived folding intermediate [29]. (b) Native state with pseudoknot. The basepairs shown in green are correctly reconstructed from the force-extension curves, see Fig. 4, using the procedure described in the main text, while the bases shown in red are involved in incorrect basepair predictions (the procedure yields no prediction for the bases shown in black); see also Fig. 5.

We will investigate whether one can in principle use the pulling-through-a-pore approach not only to discriminate between these two different conformations in individual molecules, but also to reconstruct both structures from the FEC's.

To obtain FEC's for these structures, we performed

<sup>3</sup> For instance, Liphardt *et al.* [23] observed refolding rates for a single hairpin around  $1 \text{ s}^{-1}$  at the unfolding force  $f_{1/2} \approx 14 \text{ pN}$ . At a pulling speed of say  $1 \mu\text{m/s}$ , the translocation of an RNA molecule with a thousand bases would therefore be terminated before refolding of a structural element on the *trans* side occurs.

Monte-Carlo simulations of the stochastic process defined by Eqs. (1–3), and used Eq. (6) from the Appendix to calculate the force and extension time traces. We performed all calculations at the same pulling speed ( $v = 0.1$  nm/time step, which roughly corresponds to  $10 \mu\text{m/s}$  given typical values for  $k_0$ , see above), and the same trap stiffness ( $\lambda = 0.5$  pN/nm). Fig. 4 displays three such FEC’s (solid lines) for the non-pseudoknotted structure of Fig. 3b, corresponding to unzipping from the 3’ end. These FEC’s show the sawtooth-like behavior which is characteristic for the sequential opening of structural elements (a very similar behavior was observed in the experiments of Onoa *et al.* [24] where the molecule was rapidly unzipped by pulling on its ends). The rising parts of the sawteeth correspond to stretching of single strand on the *trans* side as a stacked region is “trapped” in front of the pore on the *cis* side. When a stacked region opens, some single strand is freed to pass the pore, which leads to relaxation of the tension and causes the downstrokes in the FEC’s. Note that the FEC’s do not share all of their sawteeth, which reflects the importance of thermal fluctuations for this type of single molecule experiments (this property is manifest also in the experiment of Onoa *et al.* [24]).

The most relevant information contained in the FEC’s are the positions of the translocation arrests, during which the required force for the opening of basepairs is built up. To extract these positions, we use FEC’s of freely jointed chains with different lengths: The dashed lines in Fig. 4 show some examples of such FEC’s where the chain length  $n$  coincides with the length of the RNA single strand on the *trans* side during such an arrest. With an automated procedure described in the Appendix we obtain all of these positions (above a threshold for the duration of an arrest).

Since the bases around the position of an arrest are very likely basepaired with another segment of the sequence further to the 5’ end, we represent this information by a closing angular bracket, ‘)’, above that position in the RNA sequence (written from 5’ to 3’), see Fig. 5. Of course, the molecule can also be pulled through the pore in the other direction, i.e. from the 5’ end. This yields information on the positions of segments that have downstream binding partners. The same procedure then leads to the opening brackets, ‘(’, also shown in Fig. 5.

Bracket representations are a widely used short hand notation for RNA secondary structures. For the structures in Fig. 3, such a representation is shown in the third row of Fig. 5. Note that two types of brackets have to be used for the pseudoknotted native structure, in order to make the association between opening and closing brackets unambiguous. We observe that the angular brackets extracted from the FEC’s can be viewed as an incomplete bracket representation of the RNA secondary structure. Can we complete it using only the given sequence of the RNA molecule?

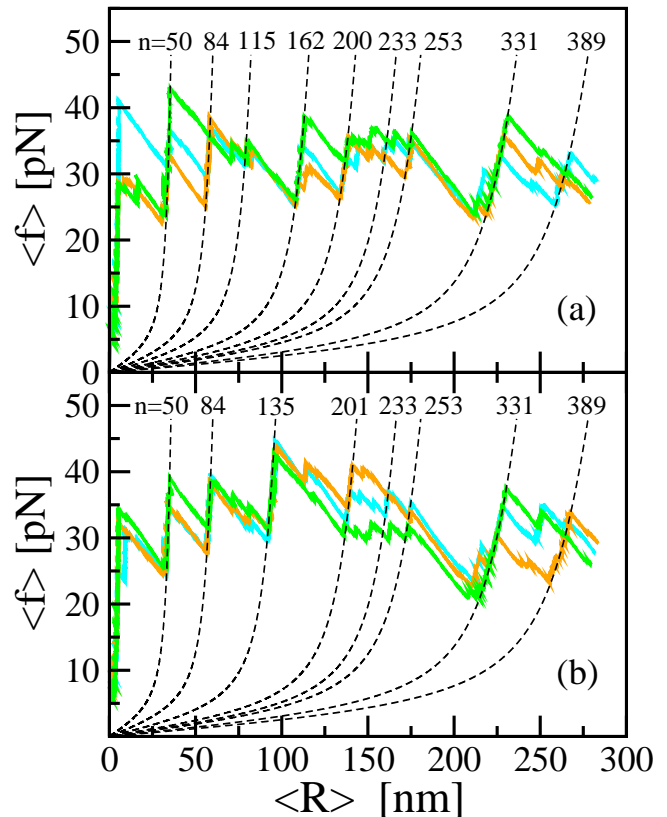


FIG. 4: Force-extension traces (solid lines) as obtained with our stochastic model for mechanical pulling through a nanopore. (a) and (b) each show three different runs with the same initial conditions (and pulling speed of  $v = 0.1$  nm/time step) for the structures in Fig. 3(a) and (b), respectively. The force  $\langle f \rangle$  and extension  $\langle R \rangle$  are calculated using Eq. (6) in the Appendix. The dashed lines are freely jointed chain FEC’s whose lengths are fitted to some of the positions that correspond to translocation arrests.

This task is a sequence alignment problem, which consists of matching each opening (closing) bracket with an associated downstream (upstream) binding sequence. Several circumstances conspire to make this, somewhat surprisingly, a nontrivial problem: (i) stems, i.e. contiguous basepaired regions, are usually short, typically 5–10 basepairs, (ii) structural elements often lead to a different number of angular brackets in the two pulling directions, i.e. not every opening bracket has a corresponding closing bracket and *vice versa*, and (iii) sequence segments containing several U’s have many possible binding partners, since U’s can pair with A’s and G’s.

To overcome this problem, we developed a probabilistic sequence alignment algorithm (see Appendix), which identifies the most likely set of stems that is consistent with all angular brackets and where all paired sequence segments contain at least one angular bracket on each side. The output of this algorithm is shown in the fourth rows of Fig. 5, where lower case letters indicate paired se-



Nanopores could in principle also be used to probe the kinetics of large-scale secondary structure rearrangements in single-molecules. For instance, it would be useful to attach larger objects to both ends of a molecule that is already threaded through the pore, allowing the same molecule to be driven forth and back through the pore, over and over again. By varying the time interval between successive reversals of the driving force, one could then probe structural relaxation over a broad range of time scales. More generally, nanopores may emerge as a new tool to probe intra- and inter-molecular interactions in single biomolecules. For instance, one could probe the biophysics of combined binding and folding in the context of RNA-protein interactions.

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## Appendix

**Calculation of free energy landscape.** Given a secondary structure of the molecule, we obtain  $\mathcal{F}_{cis}(m)$  by eliminating all basepairs involving the terminal  $N - m$  bases, and calculating the binding free energy of the remaining structure according to the free energy rules for RNA secondary structure [27]. [We take the free energy parameters as supplied with the Vienna RNA package (version 1.3.1) at room temperature  $T = 25^\circ C$ . The salt concentrations at which these parameters were measured are  $[\text{Na}^+] = 1\text{M}$  and  $[\text{Mg}^{++}] = 0\text{M}$ .] For pseudoknotted structures, the free energy rules currently include no prescription, however the following extension appears reasonable: we first eliminate basepairs in stems that give rise to the pseudoknot(s) and calculate the free energy of the remaining structure according to the standard rules. We then add the free energies of the eliminated stems separately, including the free energy for the loops created by these stems, again according to the standard free energy rules (however, the bases in these loops that are involved in other stems are removed before calculating the loop free energy).

The *trans* part of the molecule is tethered at both ends, by the pore and the pulling device, respectively. The pulling device can be described by a linear spring, while the configurational entropy of the RNA single strand can be modeled by a freely jointed chain (FJC) with extensible segments. [For the few bases that are inside the pore, we neglect the effect of the confinement on the entropy.] We denote by  $R_t$  the total extension of the *trans* part in series with the linear spring. The free energy (5) can then be expressed in terms of the total end-to-end

distance distribution  $W_{\text{FJC}+\text{spring}}$ ,

$$\mathcal{F}_{\text{FJC}+\text{spring}}(R_t; n) = -k_B T \log W_{\text{FJC}+\text{spring}}(R_t; n),$$

which can in turn be written as the convolution of the individual end-to-end distance distributions of the FJC and the spring [25],

$$W_{\text{FJC}+\text{spring}}(R_t; n) = \int_0^\infty dR W_{\text{FJC}}(R; n) W_{\text{spring}}(R_t - R).$$

Here,  $W_{\text{spring}}(R_s) = \exp(-\beta\lambda R_s^2/2)/\sqrt{2\pi/\beta\lambda}$ , where  $\lambda$  denotes the inherent stiffness of the pulling device. We calculate the end-to-end distance distribution of the freely jointed chain,  $W_{\text{FJC}}(R; n)$ , as described previously [34]. The polymer parameters we use were obtained from a fit [42] to FEC's of single-stranded DNA [41] (base-to-base length 0.7 nm, Kuhn length 1.9 nm, and stretch modulus 815 pN), since we are unaware of corresponding data for the chemically very similar RNA.

**Calculation of FEC's.** We obtain several trajectories  $m(t)$  with a Monte Carlo simulation of Eqs. (2-3) with  $m(0) = N$ ,  $R_t(0) = 0$  as initial condition and incrementing  $R_t$  at the constant rate  $v$ . The simulation is stopped when all bases have translocated ( $m = 0$ ). From the time trace  $m(t)$ , we calculate the force-extension curve  $f(R)$  using

$$\langle f \rangle = \frac{\partial}{\partial R_t} \mathcal{F}_{\text{FJC}+\text{spring}}(R_t = vt; N - m(t)) \quad (6)$$

and  $\langle R \rangle = vt - \langle f \rangle / \lambda$ . Here,  $\langle f \rangle$  and  $\langle R \rangle$  are both thermal averages over the polymer and spring degrees of freedom at fixed total extension  $R_t$  and fixed basepairing pattern.

**Extraction of parentheses positions from FEC.** For every point on a FEC, we determine the length  $n$  of the freely jointed chain whose FEC passes closest to the point (using the polymer parameters for single-stranded RNA as given above). We take a histogram of the resulting lengths  $n$  over three independent FEC's for each structure. In this histogram, the lengths  $n$  that correspond to start positions of stably basepaired regions appear as peaks, since the length of single-stranded RNA on the *trans* side remains approximately constant while the force required to unzip the basepairs builds up. [A similar procedure was applied in Ref. [43] to identify the positions of proteins bound to double-stranded DNA as it is being unzipped.] We keep all  $n$ -values where the histogram exceeds a threshold of 30 counts (a count is made every Monte Carlo time-step). Since thermal noise makes the molecule fluctuate back and forth by a few bases while the force is building up for the next stem to open, we pick out of each contiguous stretch in the remaining  $n$ -values only the largest. Finally, we increment the extracted  $n$ -values by one and mark the corresponding position in the sequence with a parenthesis.



**Reconstruction of basepairing pattern.** The FEC's do not reveal which opening and closing parentheses are paired with each other. However, given the sequence of the RNA, we can match the parentheses by sequence complementarity. [To keep the number of false basepair predictions to a minimum, we consider only stems where we have at least one parenthesis at each end.] Here, we summarize the essential steps in our sequence alignment algorithm, while a detailed presentation and characterization will be given elsewhere (R. Bundschuh and U. Gerland, to be published): First, we find all possible gapless local alignments between a subsequence containing a parenthesis and subsequences to the open side of the parenthesis, using the scoring scheme 2 for GC, 1 for AU, and 0 for GU. We keep only those alignments with a score larger than 5 and where the matching sequence segment also contains a matching parenthesis. We consider the remaining alignments as possible stems in the secondary structure. To pick the most likely set of mutually consistent stems, we assign an alignment E-value to each stem [44]. We then iteratively include the most likely stem into the structure prediction, and remove all other stems it excludes due to overlapping basepairs from the list of allowed stems.

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